Observation of Antiferromagnetic Domains in Epitaxial Thin Films

F. Nolting¹, A. Scholl¹, J. Stöhr², J. Lüning², S. Anders¹, J.W. Seo^{3,4}, J. Fompeyrine⁴, H. Siegwart⁴, J.-P. Locquet⁴, E.E. Fullerton², M.R. Scheinfein⁵, and H.A. Padmore¹ Advanced Light Source, Ernest Orlando Lawrence Berkeley National Laboratory, Berkeley, CA 94720, USA ²IBM Research Division, Almaden Research Center, 650 Harry Road, San Jose, CA 95120, USA ³Institut de Physique, University of Neuchâtel, 2000 Neuchâtel, Switzerland ⁴IBM Research Division, Zürich Research Laboratory, 8803 Rüschlikon, Switzerland ⁵Dept. of Physics and Astronomy, Arizona State University, Tempe AZ 85287-1504, USA

INTRODUCTION

The current interest in magnetic multilayers is driven by the interesting physics as well as by their application in the magnetic-storage industry [1]. An important and scientifically challenging class of materials are antiferromagnetic (AFM) thin films, because of their use in exchange bias applications [2]. Exchange bias is referred to the effect of aligning (pinning) the magnetization direction of a ferromagnet by coupling it to an AFM. This effect is routinely used in the manufacturing of advanced magnetic recording heads and it will be used in tomorrow's non-volatile magnetic memory devices. The existence of the exchange bias phenomenon has been known for more than 45 years, but the origin of this effect remains an active research area as recently reviewed by Nogués and Schuller [2]. This technologically important effect is still poorly understood because of the inability of traditional techniques to spatially determine the microscopic magnetic structure of the AFM thin film. Here we demonstrate [3] the ability of x-ray spectromicroscopy to image the magnetic structure of AFM thin films with high spatial resolution using the photoelectron emission microscope (PEEM-II) located at beam line 7.3.1.1 of the Advanced Light Source (ALS) [4].

EXPERIMENT

PEEM is a full field imaging technique where x-ray excited electrons are used to form an image of the sample surface as a function of the x-ray photon energy and polarization. This so called spectromicroscopy method combines two concepts: X-ray absorption spectroscopy and electron microscopy. Contrast can be due to a number of mechanisms including topographical, elemental, chemical, polarization, magnetic linear and circular dichroism.

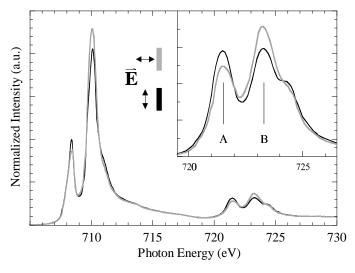


Figure 1 Total electron yield Fe L-edge XMLD spectra for a 26 nm thick film of LaFeO₃ grown on SrTiO₃(100).

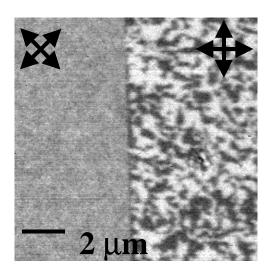


Figure 2. Image of the bycrystal junction taken with the PEEM. Arrows indicate the orientation of the crystallographic c-axis. Photon polarization E is horizontal.

The sample studied was a thin 40 nm LaFeO₃ film deposited on an asymmetric SrTiO3 (100) bicrystal where two (100) crystals were joined macroscopically at (110) and (010) faces, leading to a 45° rotation of the lattice around the surface normal. Using linearly polarized light, the angle dependent XMLD effect was measured, see Figure 1, which is due to a preferential orientation of the antiferromagnetic axis \ddot{A} . In particular, peak A at 721.5 eV is larger than peak B at 723.2 eV if the polarization of the x-ray \vec{E} is perpendicular to \vec{A} and smaller for \vec{E} parallel to it. In order to image the AFM structure of the LaFeO₃ surface we took images at 723.2 eV (peak B) and at 721.5 eV (peak A) and divided them. The resultant image (Figure 2) produces antiferromagnetic contrast and eliminates topographic contrast. The image was taken at the bicrystal junction, and reveals striking AFM domains on the right side of the junction and a uniform gray shade on the left side. The strong magnetic contrast on the right side arises from magnetic domains with an inplane projection of \ddot{A} parallel (white) and perpendicular (black) to the horizontal \vec{E} vector. On the left side the domains cannot be distinguished since \vec{E} has an equal 45° projection onto the two orientations of \vec{A} . Since in our experimental geometry \vec{E} lies in the surface plane we cannot distinguish domains whose axis \ddot{A} is rotated by 180° about the surface normal. Thus we only observe two of the four antiferromagnetic domains that have to exist by symmetry.

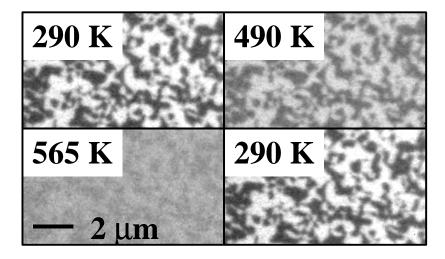


Figure 3. Temperature dependence of the antiferromagnetic domain contrast.

The temperature dependence of the XMLD effect was measured between 290 K and 550 K. The image contrast (see Figure 3) is strongly reduced at elevated temperatures and it is completely reversible upon cooling to room temperature, indication that, within the studied temperature range, the magnetic state of the sample is not affected by chemical or diffusive processes at the sample surface. By a quantitative analysis of the XMLD image contrast a Néel temperature of T_N =(670 ± 10) K was found for the film [3]. This value is reduced compared to that of the bulk LaFeO₃, with T_N = 740 K [5]. The reduced Néel temperature is unlikely due to a finite size effect. Instead, we attribute the reduced Néel Temperature to epitaxial strain.

PEEM spectromicroscopy is shown to provide the necessary resolution and contrast to resolve the generally small antiferromagnetic domains in thin films. Our studies further open the door for combined linear and circular dichrosim studies of the magnetic structure at ferromagnetic-antiferromagnetic interfaces, so-called exchange bias systems, that are both scientifically challenging and technologically important.

REFERENCES

- [1] J.B. Kortright, D.D. Awschalom, J. Stöhr, S.D. Bader, Y.U. Idzerda, S.S.P. Parkin, I.K. Schuller, and H.-C. Siegmann, J. Magn. Magn. Mater. **207**, 7 (1999).
- [2] J. Nogués and I.K. Schuller, J. Magn. Magn. Mater. **192**, 203 (1999).
- [3] A. Scholl, et al., Science, in press, (2000).
- [4] S. Anders, et al., Rev. Sci. Instrum. **70**, 3973 (1999).
- [5] M. Eibschütz, S. Shtrikman, and D. Treves, Phys. Rev. **156**, 562 (1967).

This work was supported by the Division of Chemical Sciences (SSRL) and the Division of Materials Science (ALS) of the Office of Basic Energy Sciences of the U.S. Department of Energy. J.W.S and F.N. acknowledge support by the Swiss National Science Foundation.

Principal investigator: Joachim Stöhr, IBM Research Division, Almaden Research Center. Email: stohr@ssrl.slac.stanford.edu. Telephone: 650-926-2570.